

Classical and quantum ordering of protons in cold solid hydrogen under megabar pressures

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A combination of state-of-the-art theoretical methods has been used to obtain an atomic-level picture of classical and quantum ordering of protons in cold high-pressure solid hydrogen. We focus mostly on phases II and III of hydrogen, exploring the effects of quantum nuclear motion on certain features of these phases (through a number of *ab initio* path integral molecular dynamics (PIMD) simulations at particular points on the phase diagram). We also examine the importance of van der Waals forces in this system by performing calculations using the optB88-vdW density functional, which accounts for non-local correlations. Our calculations reveal that the transition between phases I and II is strongly quantum in nature, resulting from a competition between anisotropic inter-molecular interactions that restrict molecular rotation and thermal plus quantum fluctuations of the nuclear positions that facilitate it. The transition from phase II to III is more classical because quantum nuclear motion plays only a secondary role and the transition is determined primarily by the underlying potential energy surface. A structure of $P2_1/c$ symmetry with 24 atoms in the primitive unit cell is found to be stable when anharmonic quantum nuclear vibrational motion is included at finite temperatures using the PIMD method. This structure gives a good account of the infra-red (IR) and Raman vibron frequencies of phase II. We find additional support for a $C2/c$ structure as a strong candidate for phase III, since it remains transparent up to 300 GPa, even when quantum nuclear effects are included. Finally, we find that accounting for van der Waals forces improves the agreement between experiment and theory for the parts of the phase diagram considered, when compared to previous work which employed the widely-used Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional.

I. INTRODUCTION

Hydrogen (H) is the most abundant of the elements and, having one electron, it can form only a single strong covalent bond. As a result, pure H is expected to remain molecular to very high pressures. Several solid phases of H have been observed; phase I is a quantum crystal of rotating H₂ molecules on a hexagonal close packed (hcp) lattice but, despite decades of study, the arrangements of the molecules in phases II and III are unknown.^{1,2} Evidence has recently been found for a room-temperature phase IV of solid hydrogen,³⁻⁶ although here we study only low temperatures of $T \leq 150$ K.

Determining the atomic structures of the different phases presents a formidable challenge to experiment. Direct experimental structure determinations for solid hydrogen are hampered by the weak scattering of X-rays by H and the difficulties of combining high-pressure static diamond anvil cell techniques with such measurements. To date only a single X-ray study covering the three low-temperature phases has been reported.⁷ Although this study provides constraints on the experimental structures and phase transitions, the strongest experimental evidence comes from IR, Raman, and optical measurements which provide valuable but indirect structural information.¹ A key experimental observation is that the boundary between phases I and II depends strongly on the isotope.^{1,8,9} At low temperatures, the I/II transition occurs at 28 to 40 GPa (0.28 to 0.4 megabar) in D₂,^{9,10} 50 to 69 GPa in HD,¹¹ and 90 to 110 GPa in H₂.^{8,12} In contrast, the boundary between phases II and III, which occurs at ~ 160 GPa,¹³ is only weakly dependent on the isotope. Likewise, the transition from phase I to II is strongly temperature dependent, whereas the II/III transition is essentially independent of temperature. Upon passing from phase I to II the vibrational roton modes undergo substantial changes, indicating that the molecules change from a rotationally free state to a rotationally restricted one. The IR activity increases dramatically at the II/III transition and the IR and Raman vibron frequencies soften by about 80 cm^{-1} .^{7,14} Phase III remains transparent up to 300 GPa but becomes dark by about 320 GPa.¹⁴

The theoretical characterisation of high pressure solid hydrogen is also a major challenge. The development of a successful model must start with an extensive exploration of structural phase space in order to identify candidate structures for the different phases.^{15,16} In addition, the energetics of the structures must be accurately accounted for, and the optical properties must be well-described, particularly at the high pressures at which metallisation may occur. Given the low mass of hydrogen, quantum nuclear effects including zero-point (ZP) motion should also be accounted for.¹⁷⁻²⁰ No current theoretical method is capable of giving a satisfactory account of all of these, and none of

the excellent theoretical papers on this system have addressed all of these issues simultaneously. In this work we have addressed these points using a combination of state-of-the-art methods. We have used the results from extensive density-functional-theory (DFT) based searches of the potential energy surface (PES)^{15,21}, so that our simulations start from the most stable structures identified to date. We have used the *ab initio* path-integral molecular dynamics (PIMD) method^{22–24} to account for quantum nuclear motion at finite temperatures. The equilibrated structures obtained from the PIMD calculations are used to calculate optical properties using the *GW* many-body perturbation theory approach, and a hybrid exchange-correlation density functional.²⁵ In the PIMD simulations we use a 144-atom supercell together with the constant-pressure, constant-temperature (NPT) ensemble.²⁶ This ensemble allows relaxation of the cell shape and size, and reduces the bias towards particular structures which arises from using small fixed cells. The use of large variable cells and lower-enthalpy static-lattice structures as the starting points of our simulations are important advances over previous PIMD studies of solid H, where either “quantum localisation” or metallic structures at too low pressures were predicted.^{18–20} As we show below, the broad range of methods employed here provides an atomic-level picture of the classical and quantum ordering of protons in low-temperature solid hydrogen that is consistent with the key experimental results.

II. COMPUTATIONAL DETAILS

The quantum behaviour of the nuclei is described within the PIMD method by a “ring polymer” consisting of beads connected by springs.^{22–24} Nuclear exchange effects are neglected in our calculations and we assume an adiabatic decoupling of the electron and nuclear motions. PIMD is a finite temperature method and the number of beads required to give an accurate sampling of the imaginary time path decreases with increasing temperature. We have used 64 beads per nucleus for the simulations at 50 and 100 K, and 16 beads at 150 K. We have performed NPT PIMD simulations at 80 and 200 GPa. The simulations at 80 GPa were run for 9,000 steps using a time step of 0.25 fs, and the final 8,000 steps were included in the data analysis. Those at 200 GPa were run for 20,000 steps with a time step of 0.125 fs, and data from the final 15,000 steps were included in the analysis. Convergence tests were performed on the number of beads used, the simulation length, and the sampling intervals, and we also tested the validity of neglecting the nuclear exchange in the PIMD calculations, as reported in the supporting information. The DFT-based classical molecular dynamics (MD) and PIMD simulations were of the Born-Oppenheimer type, *i.e.*, the electronic density was optimised self-consistently to the ground state at each time step. The electronic states

were sampled on a k-point grid of spacing $0.05 \times 2\pi/\text{\AA}$ in all of the calculations. The MD and PIMD simulations were carried out using the CASTEP code²⁷ with ultra-soft pseudopotentials²⁸ and a 350 eV plane-wave cut-off. The IR and Raman spectra were calculated within density-functional perturbation theory²⁹, using the CASTEP code, norm-conserving pseudopotentials^{30,31} and a plane-wave cut-off energy of 800 eV. All of the CASTEP calculations were performed with the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional³². The vdW-DF (optB88-vdW)³³, *GW*, and hybrid functional calculations were carried out using the VASP^{34,35} code with PAW potentials^{36,37} and a 600 eV plane-wave cut-off. The X-ray diffraction simulations were performed using the Powder Cell program.³⁸

III. RESULTS AND DISCUSSIONS

A. Accuracy of PBE

Pickard and Needs performed extensive explorations of the structural phase space of hydrogen, and a number of low-enthalpy structures were identified at pressures of about 100 GPa where phase II is stable.^{15,21} These structures correspond to minima in the static lattice enthalpy, and they do not include any effects from vibrations. The molecular centres in these structures lie very close to a hcp lattice, as is generally thought to occur in phase II. In this study we have used the $P2_1/c$ -24 structure²¹ as a model for phase II. The $P2_1/c$ -24 structure has 24 atoms in the primitive unit cell²¹, and it is a little more stable than the structure of the same symmetry reported in Ref. 25, which has an 8-atom primitive unit cell. At higher pressures these hcp-based structures become unstable with respect to a layered structure of $C2/c$ symmetry with 24 atoms in the primitive cell. A related and slightly less stable structure, with the same space group but a smaller unit cell, was found by Tse *et al.*¹⁶. DFT calculations using the PBE exchange-correlation functional show that $C2/c$ has the lowest static-lattice enthalpy among the plausible candidates for phase III in most of the pressure range in which it has been observed, see Fig. 1 and Refs. 15 and 21. Although PBE has been successfully applied in many high pressure studies, it does not account for van der Waals (vdW) interactions, which are often important in binding molecular crystals. We have therefore tested the effects of including vdW forces on the relative enthalpies and phase boundaries using the optB88-vdW functional³³ within the vdW-DF scheme.³⁹ Fig. 1 shows that the relative stabilities of the structures are generally not altered substantially. However, the transition pressure between phases II and III (taking the $P2_1/c$ -24 structure as our model for phase II and

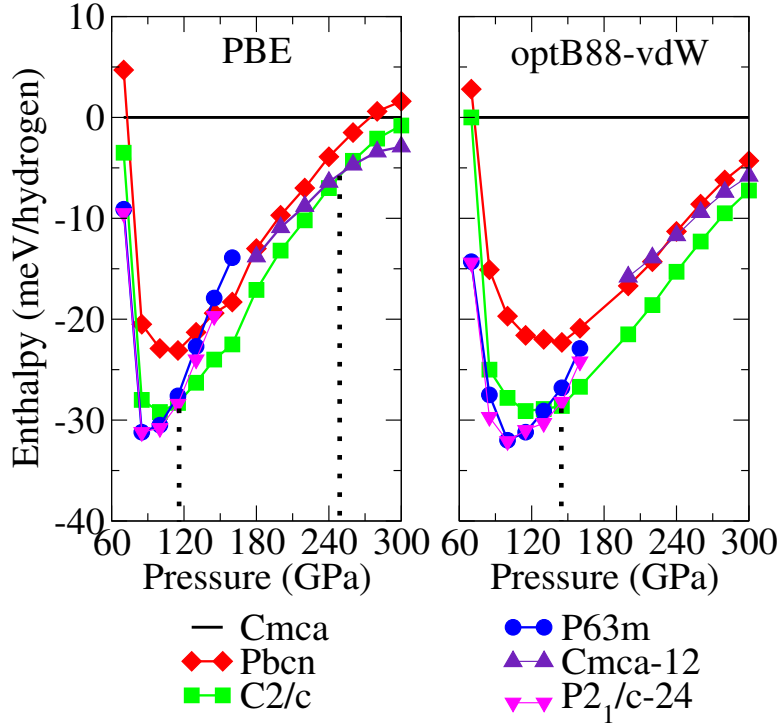


FIG. 1. Enthalpy as a function of pressure relative to the $Cmca$ structure²⁵ at megabar pressures. The results shown in the left panel were calculated using the PBE functional and those in the right panel were obtained using the optB88-vdW functional, which accounts for van der Waals (vdW) interactions within the vdW-DF scheme. The $Cmca$ -12 molecular structure is described in Ref. 15.

$C2/c$ for phase III) is increased from ~ 120 GPa with PBE to ~ 150 GPa with optB88-vdW, in better agreement with experiment.¹³ As can be seen in Fig. 1, static lattice PBE calculations indicate that $Cmca$ ²⁵ is less favourable than $C2/c$ over the pressure range in which phase III is observed. However, including harmonic ZP motion changes the picture significantly^{5,6}, and the softer $Cmca$ structure becomes the most energetically favourable above about 225 GPa. The $Cmca$ structure is, however, metallic and its vibronic frequencies are considerably lower than those of phase III, and therefore $Cmca$ is not a plausible model for phase III. As can be seen from Fig. 1, using the optB88-vdW functional³³ instead of PBE leads to an increase in the enthalpies with respect to $C2/c$ of about 6 meV per proton for the weakly molecular $Cmca$ structure and about 3 meV per proton for the strongly molecular $Cmca$ -12 structure. When harmonic ZP motion corrections are added to the static lattice data of Fig. 1 we find that $C2/c$ remains the most stable phase up to about 300 GPa. Using the optB88-vdW functional therefore removes a major conflict with experiment in the PBE phase diagram reported in Refs. 5 and 6 and, overall, gives significantly better agreement with the experimental phase diagram than the PBE functional.

B. Phases I and II

In order to explore phases I and II in more detail, we have carried out an extensive series of *ab initio* PIMD simulations. The first set of calculations was performed at 80 GPa, a pressure safely within the experimental region of stability of phase II in solid D₂ and the region in which static DFT calculations predict phase II-like structures to be stable. PIMD simulations were performed on both the $P6_3/m$ and $P2_1/c-24$ structures since they are the two lowest-enthalpy static-lattice structures at 80 GPa. A 144-atom supercell was used which is commensurate with the cells of both the $P6_3/m$ ¹⁵ and $P2_1/c-24$ structures.²¹ Using D at a temperature of 50 K and starting from the $P6_3/m$ structure, MD and PIMD simulations quickly showed a transformation from the original structure to a new rotationally restricted state. The centres of the molecules still form a hcp lattice but the molecular orientations differ from the starting state and it is difficult to identify a corresponding static structure. This indicates that although $P6_3/m$ corresponds to a minimum in the PES (with positive harmonic vibrational frequencies), it is so shallow that thermal or quantum effects are sufficient to disrupt it. Therefore $P6_3/m$ is not a plausible structure for phase II when anharmonic nuclear motion is taken into account.

Starting from the $P2_1/c-24$ structure, we performed a classical MD simulation at 50 K, PIMD runs for D and H at 50 K, and a PIMD run for D at 150 K. Some of the key results from these simulations are reported in Fig. 2, where representative configurations of the centroid trajectories are plotted. We find that the orientational ordering of the starting structure is maintained in the classical MD simulation (Fig. 2(a) and (b)). Similarly, the initial orientational ordering is rather well preserved in the PIMD simulation with D (Fig. 2 (c) and (d)). Under the same conditions, but switching to the lighter H atom, we find that the orientational ordering is almost completely lost, as reflected by the spherical distributions seen in Fig. 2(e) and (f). Returning to the heavier D atom but increasing the temperature to 150 K, we again find a spherical trajectory distribution on a hcp lattice. These qualitative interpretations are supported by a detailed analysis of the probability distributions associated with the rotational freedom of the molecules. We calculate the angle θ between the projection of the molecules on the $x - y$ plane and the y axis (inset of Fig. 3 (d)), and plot probability distributions of θ and the molecular bond length (r) for each simulation. The geometry-optimised $P2_1/c-24$ structure has molecules aligned between 30° and 50° and between -30° and -50°. Two rather broad peaks are seen in the PIMD simulations for D at 50 K at about $\pm 40^\circ$ (Fig. 3 (b)), which are consistent with restricted molecular rotation. On the other hand, either replacing D by H or raising the temperature to 150 K destroys the peak structure and leads

to a probability distribution characteristic of a freely rotating molecular state (Fig. 3 (c) and (d)). These results show that the degree of orientational order is governed by a competition between the corrugation of the underlying PES, which favours rotational restriction, and the thermal and quantum nuclear fluctuations, which oppose it. We find no evidence for the “quantum localisation” reported in previous PIMD simulations.²⁰

Our results explain in a simple manner the positive slope (dP/dT) of the I/II coexistence line as well as the isotope-dependence of the transition pressure. The corrugation of the underlying PES increases with pressure and, as the pressure increases, higher temperatures are required to overcome the corrugation. The I/II transition for hydrogen occurs at a higher pressure than in deuterium for a similar reason; quantum effects are greater for hydrogen and so the corrugation of the underlying PES is more easily overcome by hydrogen than by deuterium. These results are consistent with the traditional understanding of the quantum nature of phase I.^{1,8}

C. Phase III

DFT structure searches⁴⁰ have so far provided a single best candidate for phase III: a layered $C2/c$ structure with significant deviations from close packing within the layers.¹⁵ Other competitive structures were found, including an intriguing structure of $Pbcn$ symmetry which consists of alternate layers of strongly bonded and more weakly bonded molecules (see Fig. 3 of the supplementary information for Ref. 15). $Pbcn$ has a lower ZP enthalpy than structures consisting entirely of strongly bonded molecules, and it might be favoured when ZP motion is taken into account. We therefore performed a number of simulations for both $Pbcn$ and $C2/c$. In particular, we focused on understanding whether either of these structures is consistent with the experimental observation that phase III is transparent up to 300 GPa, but becomes opaque by 320 GPa.¹⁴ To this end we calculated the band gap energies of the $Pbcn$ and $C2/c$ structures using a many-body perturbation theory approach and the GW approximation to the self energy, and PBE0-like density functionals with different percentages of “exact-exchange” (EXX).⁴¹ We found that using a hybrid density functional including 30% EXX reproduced the direct and lowest band gaps of the $Pbcn$ and $C2/c$ structures obtained within GW theory (Fig. 4). This hybrid density functional was subsequently used to calculate the optical absorption spectra of the static lattice $Pbcn$ and $C2/c$ structures at pressures between 200 and 350 GPa, see Fig. 5(a) and (b). The peaks in the absorption spectra correspond to strong inter-band transitions, and a non-zero imaginary part of the macroscopic dielectric function in the visible light range indicates that the material is to some extent opaque. The

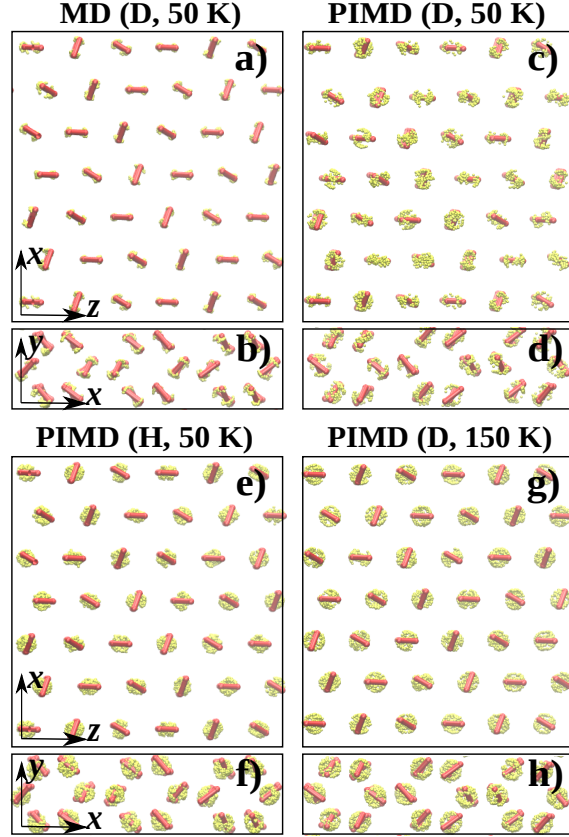


FIG. 2. Trajectories of structures obtained from simulations with classical and quantum nuclei at 80 GPa starting from the $P2_1/c$ -24 structure. Yellow balls show the representative configurations of the centroids throughout the course of the simulation. The red rods show the static (geometry-optimised) structure. A conventional hexagonal cell containing 144 atoms was used. Panels a, c, e, and g show the $z-x$ plane and panels b, d, f, and h show the $x-y$ plane of the hcp lattice. The four simulations are: 1) MD with classical nuclei at 50 K (panels a and b), 2) PIMD for D at 50 K (panels c and d), 3) PIMD for H at 50 K (panels e and f), and 4) PIMD for D at 150 K (panels g and h). In the MD simulation, the anisotropic inter-molecular interaction outweighs the thermal and quantum nuclear fluctuations. Therefore, the molecular rotation is highly restricted. The thermal plus quantum nuclear fluctuations outweigh the anisotropic inter-molecular interactions in the PIMD simulations of H at 50 K and D at 150 K.

evolution of the absorption spectra shows that $Pbcn$ becomes opaque between 270 and 300 GPa (Fig. 5a), while $C2/c$ turns opaque between 300 and 350 GPa (Fig. 5b). Considering the limited accuracy of the static pressure, the static-lattice absorption spectra of both $Pbcn$ and $C2/c$ are in qualitative agreement with the experimental observations.

We now consider the changes in the absorption spectra induced by introducing thermal and quantum nuclear fluctuations. To this end we have performed PIMD simulations for H using 48-atom cells, starting from the $Pbcn$ and $C2/c$ structures, at 100 K and 200 GPa, a pressure

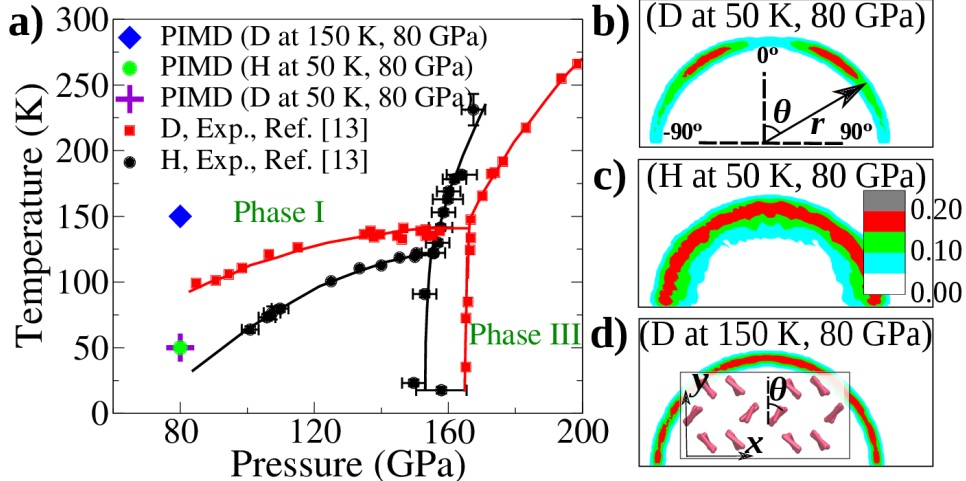


FIG. 3. Phase diagram of solid H and D with the pressure-temperature coordinates of three key PIMD simulations indicated at 80 GPa (in panel a), and probability distribution functions for these simulations (panels b, c, and d). The three PIMD simulations are: 1) D at 50 K, 2) H at 50 K, and 3) D at 150 K. D at 50 K has the least thermal plus quantum nuclear fluctuations, while H at 50 K and D at 150 K have larger thermal plus quantum fluctuations. The probability is plotted as a function of r (molecular bond length) and θ (angle between the projection of the molecules on the $x - y$ plane and the y axis). The ground state structure has molecules aligned between 30° and 50° and between -30° and -50° (inset in panel d and Fig. S1 of the supporting information). The molecules of D at 50 K (panel b) are still rotationally restricted (phase II). Replacing D by H (panel c) or elevating the temperature to 150 K (panel d) leads to free rotation (phase I). Thermal plus quantum fluctuations compete with the anisotropic inter-molecular interactions in the region of the transition from phases I to II.

well within the stability range of phase III. These reveal that the $Pbcn$ structure maintains its original symmetry but the weak molecular layers evolve to atomic-like layers in which the nearest neighbour distances are approximately equal. When using the new PIMD structure to calculate the absorption spectra we find that $Pbcn$ is already opaque at 200 GPa (note the absorption peak within the visible light region in Fig. 5(c)). This is inconsistent with experiments on phase III which find that it remains transparent up to ~ 300 GPa. However, in the PIMD simulations for the $C2/c$ phase, the structure and consequently the absorption spectrum are only slightly altered compared with the original geometry-optimised structure (Fig. 5(d)). Therefore, unlike $Pbcn$, $C2/c$ remains transparent up to pressures as high as 300 GPa when thermal and quantum fluctuations are taken into account. Zha *et al.*⁴² recently reported synchrotron IR measurements for dense hydrogen, finding that the IR vibron of phase III persists up to at least 360 GPa. This behaviour is also reasonably consistent with the $C2/c$ structure.

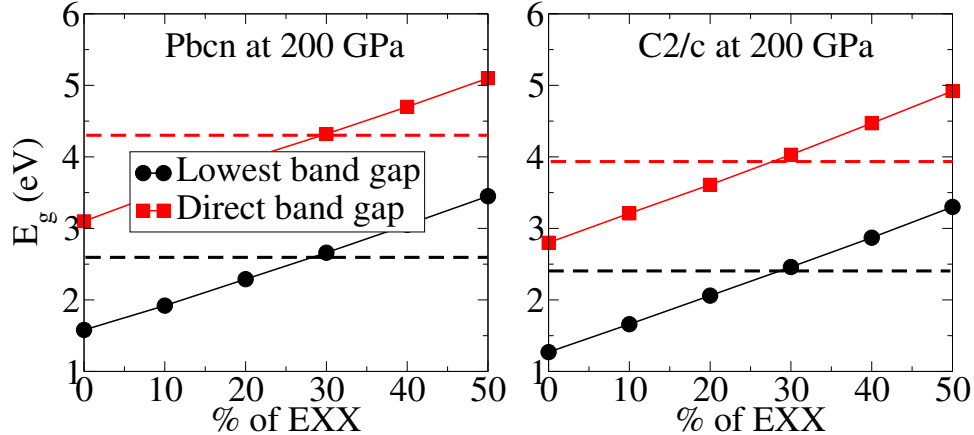


FIG. 4. Variation of the lowest and direct band gaps with the percentage of exact-exchange (EXX) in the hybrid functional calculations for the static *Pbcn* (left panel) and *C2/c* (right panel) structures at 200 GPa. The black (red) dashed line indicates the LDA based G_0W_0 results for the lowest (direct) band gap. The G_0W_0 results are converged to within 0.01 eV with respect to the number of unoccupied states. Using 30% exact-exchange reproduces the G_0W_0 band gaps to within ~ 0.1 eV.

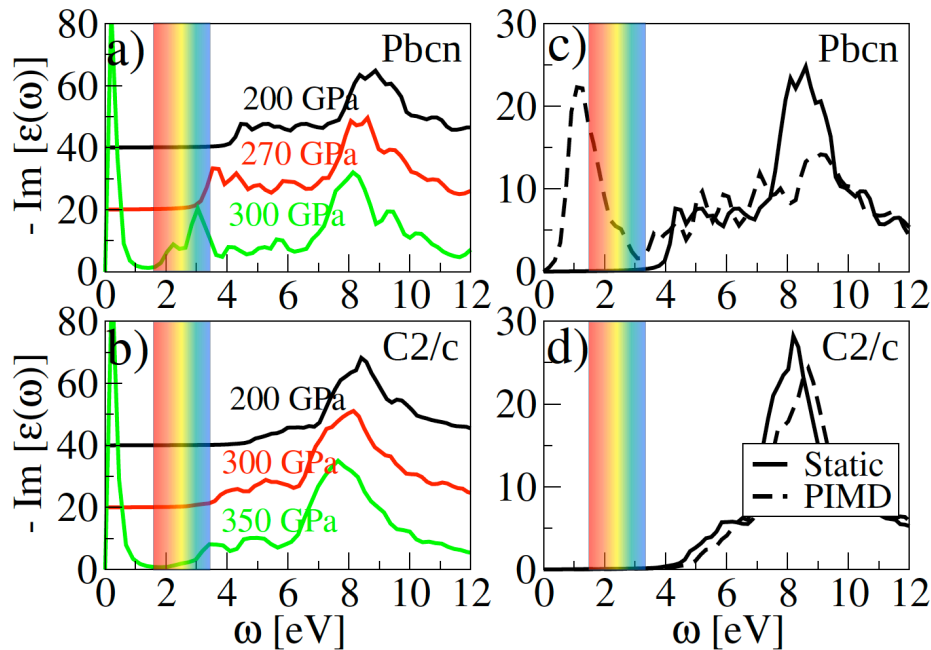


FIG. 5. Absorption spectra of hydrogen at 200 GPa. Panels (a) and (b) show absorption spectra calculated using 30% exact exchange at different pressures for (a) *Pbcn*, and (b) *C2/c*, using static geometry-optimised structures. Panels (c) (*Pbcn*) and (d) (*C2/c*) compare the 200 GPa absorption spectra obtained from static geometry-optimised structures (solid lines) with those obtained from the PIMD simulations (dashed lines). The visible light region is shown by the rainbow in each panel. Using the PIMD structure renders *Pbcn* opaque at 200 GPa, but has little effect on *C2/c*.

D. Discussion and Conclusions

From the previous discussion it is clear that our most promising candidate structure for phase II is $P2_1/c-24$, but we have insufficient evidence to identify it as the correct structure. The $P2_1/c-24$ structure is monoclinic, while experiments have suggested that phase II is hexagonal.⁴³ Experiments on D_2 have suggested that phase II is incommensurate⁴³, and in a theoretical study¹⁸ it was speculated that phase II of H_2 might be “diffuse”, meaning that it does not correspond to a single classical structure. We assign space group symmetries to the static lattice structures, not to the structures found in the PIMD simulations which include nuclear motion. It is possible that the nuclear motion might reduce lattice distortions, similar to the way in which heating can lead to more symmetric structures, although such an effect was not apparent in our PIMD simulations. The differences between the results obtained with the PBE and optB88-vdW functionals indicate some sensitivity to the functional used, which must temper our confidence in the accuracy of DFT results for this system. In addition, our simulations do not account for nuclear exchange effects such as those that occur in *ortho* and *para* molecules, which are known to be important in a quantitative description of phase I and its transition to phase II^{44–47}. Notwithstanding these limitations, it is clear that quantum nuclear effects and molecular rotations play a very important role in the transition between phases I and II and the transition is strongly quantum in nature even when nuclear exchange is neglected.

As the pressure increases the molecules rearrange with a significant distortion from close packing, so that the molecules can avoid one another. The difference in the thermal plus quantum motion of the two phases plays only a secondary role and the II/III transition is therefore only weakly dependent on the isotope and temperature. Based on the observation of persistent (100) and (101) X-ray diffraction peaks up to 180 GPa, Akahama *et al.* have concluded that any distortion from hcp packing in phase III is small.⁷ This appears to rule out the $C2/c$ structure. However, we have calculated X-ray diffraction data assuming $P2_1/c-24$ for phase II and $C2/c$ for phase III, finding the positions of the (100) and (101) peaks in the hcp lattice to be in very good agreement with experiment over a wide pressure range,⁷ see Fig. 6(c). We have also found the variation with pressure of the c/a ratios of $P2_1/c-24$ and $C2/c$ to be in good agreement with experiment, including the discontinuous drop at the II/III transition (Fig. 6(d)).⁷ The variations with pressure of the IR and Raman frequencies are well accounted for by the $P2_1/c-24$ and $C2/c$ structures, see Fig. 6(a) and (b). The two ends of a molecule in the $C2/c$ structure have different environments, so the molecules have dipole moments and the structure shows intense IR vibron activity (Fig. 6(a)

inset), as found in experiments on phase III.

In conclusion, we have studied solid hydrogen at megabar pressures using a variety of *ab initio* techniques. The transition from phase I to II is governed by a competition between the corrugation of the underlying PES, which restricts molecular rotation, and the thermal and quantum nuclear fluctuations, which facilitate it. At very low pressures and temperatures the ZP motion washes out the corrugation of the PES and leads to the freely-rotating phase I for both isotopes. Including ZP motion destabilises the $P6_3/m$ structure, which is therefore not a plausible candidate for phase II. Our simulations favour the proposal that phase II is a hcp-based rotationally restricted molecular phase with a large cell.⁴³ The IR and Raman active vibron frequencies of the $P2_1/c-24$ structure are in good agreement with the experimental data for phase II.⁴⁸ We also find that the (100) and (101) X-ray diffraction peak positions of $P2_1/c-24$ and $C2/c$, and their c/a ratios, are in good agreement with experiment.⁷ The optB88-vdW³³ density functional gives a II/III transition pressure in much closer agreement with experiment than PBE, and it moves the $Cmca$ and $Cmca-12$ phases to higher enthalpies, so that the $C2/c$ structure becomes the most stable phase up to about 300 GPa. Using the optB88-vdW functional therefore removes a major discrepancy between the theoretical and experimental phase diagrams. We have found that $C2/c$ remains transparent up to 300 GPa, which provides further support for it as a model for phase III. Overall, our study provides evidence from a range of state-of-the-art theoretical methods which supports the picture of orientational ordering proposed by Mazin *et al.*⁸ and others. We have provided an atomic-level picture of the evolution of solid hydrogen under megabar pressures which satisfactorily explains some of the experimental results, and can be tested by further experimental and theoretical studies.

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¹ H. K. Mao and R. J. Hemley, Rev. Mod. Phys. **66**, 671 (1994)

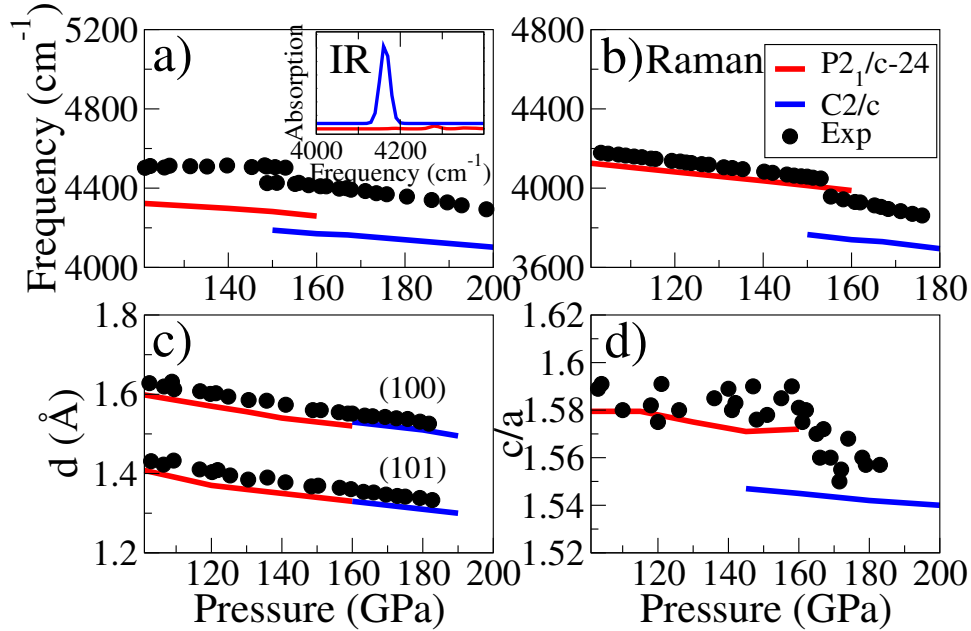


FIG. 6. Comparison of calculated and experimental properties of phases II and III, assuming the $P2_1/c-24$ structure for phase II and $C2/c$ for phase III. Variation with pressure of the IR (a) and Raman (b) vibron frequencies. The inset in (a) shows the dramatic difference in IR intensity between $P2_1/c-24$ (phase II) and $C2/c$ (phase III). The $C2/c$ spectrum is slightly shifted to distinguish it from that of $P2_1/c-24$. (c) Variation with pressure of the (100) and (101) X-ray diffraction peak positions. (d) Variation with pressure of the c/a ratio. Experimental data is indicated by circles; the data in (a) and (b) were taken from Ref. 48 and those in (c) and (d) were taken from Ref. 7.

- ² J. M. McMahon, M. A. Morales, C. Pierleoni, and D. M. Ceperley, *Rev. Mod. Phys.* **84**, 1607 (2012)
- ³ M. I. Eremets and I. A. Troyan, *Nature Mater.* **10**, 927 (2011)
- ⁴ R. T. Howie, C. L. Guillaume, T. Scheler, A. F. Goncharov, and E. Gregoryanz, *Phys. Rev. Lett.* **108**, 125501 (2012)
- ⁵ C. J. Pickard, M. Martinez-Canales, and R. J. Needs, *Phys. Rev. B* **85**, 214114 (2012)
- ⁶ C. J. Pickard, M. Martinez-Canales, and R. J. Needs, *Phys. Rev. B* **86**, 059902(E) (2012)
- ⁷ Y. Akahama, M. Nishimura, H. Kawamura, N. Hirao, Y. Ohishi, and K. Takemura, *Phys. Rev. B* **82**, 060101 (2010)
- ⁸ I. I. Mazin, R. J. Hemley, A. F. Goncharov, M. Hanfland, and H. K. Mao, *Phys. Rev. Lett.* **78**, 1066 (1997)
- ⁹ L. J. Cui, N. H. Chen, S. J. Jeon, and I. F. Silvera, *Phys. Rev. Lett.* **72**, 3048 (1994)
- ¹⁰ I. F. Silvera and R. J. Wijngaarden, *Phys. Rev. Lett.* **47**, 39 (1981)
- ¹¹ F. Moshary, N. H. Chen, and I. F. Silvera, *Phys. Rev. Lett.* **71**, 3814 (1993)
- ¹² H. E. Lorenzana, I. F. Silvera, and K. A. Goettel, *Phys. Rev. Lett.* **64**, 1939 (1990)
- ¹³ A. F. Goncharov, R. J. Hemley, and H. K. Mao, *J. Chem. Phys.* **134**, 174501 (2011)

- ¹⁴ P. Loubeyre, F. Occelli, and R. LeToullec, *Nature* **416**, 613 (2002)
- ¹⁵ C. J. Pickard and R. J. Needs, *Nature Phys.* **3**, 473 (2007)
- ¹⁶ J. S. Tse, D. D. Klug, Y. Yao, Y. L. Page, and J. R. Rodgers, *Solid State Commun.* **145**, 5 (2008)
- ¹⁷ J. Kohanoff, S. Scandolo, G. L. Chiarotti, and E. Tosatti, *Phys. Rev. Lett.* **78**, 2783 (1997)
- ¹⁸ S. Biermann, D. Hohl, and D. Marx, *J. Low Temp.* **110**, 97 (1998)
- ¹⁹ S. Biermann, D. Hohl, and D. Marx, *Solid State Commun.* **108**, 337 (1998)
- ²⁰ H. Kitamura, S. Tsuneyuki, T. Ogitsu, and T. Miyake, *Nature* **404**, 259 (2000)
- ²¹ C. J. Pickard and R. J. Needs, *Phys. Status Solidi (b)* **246**, 536 (2009)
- ²² D. Marx and M. Parrinello, *Z. Phys. B (Rapid Note)* **95**, 143 (1994)
- ²³ D. Marx and M. Parrinello, *J. Chem. Phys.* **104**, 4077 (1996)
- ²⁴ M. E. Tuckerman, D. Marx, M. L. Klein, and M. Parrinello, *J. Chem. Phys.* **104**, 5579 (1996)
- ²⁵ K. A. Johnson and N. W. Ashcroft, *Nature* **403**, 632 (2000)
- ²⁶ G. J. Martyna, A. Hughes, and M. E. Tuckerman, *J. Chem. Phys.* **110**, 3275 (1999)
- ²⁷ S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. J. Probert, K. Refson, and M. C. Payne, *Z. Kristallographie* **220**, 567 (2005)
- ²⁸ D. Vanderbilt, *Phys. Rev. B* **41**, 7892 (1990)
- ²⁹ S. Baroni, S. de Gironcoli, A. D. Corso, and P. Giannozzi, *Rev. Mod. Phys.* **73**, 515 (2001)
- ³⁰ D. R. Hamann, *Phys. Rev. B* **40**, 2980 (1989)
- ³¹ N. Troullier and J. L. Martins, *Phys. Rev. B* **43**, 1993 (1991)
- ³² J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996)
- ³³ J. Klimeš, D. R. Bowler, and A. Michaelides, *J. Phys.: Condens. Matter* **22**, 022201 (2010)
- ³⁴ G. Kresse and J. Furthmüller, *Comput. Mat. Sci.* **6**, 15 (1996)
- ³⁵ J. Klimeš, D. R. Bowler, and A. Michaelides, *Phys. Rev. B* **83**, 195131 (2011)
- ³⁶ P. E. Blöchl, *Phys. Rev. B* **50**, 17953 (1994)
- ³⁷ G. Kresse and D. Joubert, *Phys. Rev. B* **59**, 1758 (1999)
- ³⁸ W. Kraus and G. Nolze, *Journal of Applied Crystallography* **29**, 301 (1996)
- ³⁹ M. Dion, H. Rydberg, E. Schroder, D. C. Langreth, and B. I. Lundqvist, *Phys. Rev. Lett.* **92**, 246401 (2004)
- ⁴⁰ C. J. Pickard and R. J. Needs, *J. Phys.: Condens. Matter* **23**, 053201 (2011)
- ⁴¹ C. Adamo and V. Barone, *J. Chem. Phys.* **110**, 6158 (1999)
- ⁴² C. S. Zha, Z. X. Liu, and R. J. Hemley, *Phys. Rev. Lett.* **108**, 146402 (2012)
- ⁴³ I. Goncharenko and P. Loubeyre, *Nature* **435**, 1206 (2005)
- ⁴⁴ J. Igarashi, *J. Phys. Soc. Jap.* **59**, 2811 (1990)
- ⁴⁵ A. F. Goncharov, M. A. Strzhemechny, H. K. Mao, and R. J. Hemley, *Phys. Rev. B* **63**, 064304 (2001)
- ⁴⁶ Y. A. Freiman, S. Tretyak, A. Jezowski, and R. J. Hemley, *J. Low Temp. Phys.* **122**, 537 (2001)
- ⁴⁷ Y. A. Freiman, S. Tretyak, A. F. Goncharov, H. K. Mao, and R. J. Hemley, *Low Temp. Phys.* **37**, 1038 (2011)

⁴⁸ M. Hanfland, R. J. Hemley, and H. K. Mao, Phys. Rev. Lett. **70**, 3760 (1993)